

Full Paper

One-step Electrophoretic/electrochemical Synthesis of Reduced Graphene Oxide/Manganese Oxide (rGO-Mn₃O₄) Nanocomposite and Study of its Capacitive Performance

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Abstract- Herein, a very simple and scalable electrochemical strategy is reported to fabricate reduced graphene oxide (rGO)-Mn₃O₄ nanoparticles composite for supercapacitors. First, graphene oxide is directly produced via Hummers method, and dispersed in manganese nitrate aqueous solution. The rGO-Mn₃O₄ particles nanocomposite is then prepared through electrochemically decoration of Mn₃O₄ nanoparticles onto reduced graphene oxide (rGO) sheets. The formation of composite is explained by electrophoretic/electrochemical deposition (EPD/ECD) mechanism. The as-prepared rGO-Mn₃O₄ nanocomposite is characterized by XRD, IR, FE-SEM and TEM techniques. These analyses results specified the co-deposition of manganese oxide particles and rGO sheets on the cathode electrode. The charge storage ability of the fabricated nanocomposite as electrode material for supercapacitors was studied using cyclic voltammetry and charge-discharge techniques. The obtained electrochemical data indicated that the rGO-Mn₃O₄ nanocomposite enable to provide specific capacitance of 364 F g⁻¹ at 2 A g⁻¹ and 86.3% of its initial capacity after 5000 charge/discharging.

Keywords- rGO-Mn₃O₄ nanocomposite, Nanoparticles, Graphene oxide, Electrochemical synthesis

1. INTRODUCTION

Supercapacitors (SCs) have recently gained significant attention due to their excellent charge–discharge performance, long-term cycle life, and high power densities [1]. Furthermore, SCs could also make up the difference in energy and power between batteries and traditional capacitors. It was reported that transition metal oxides (TMOs) will play a significant role in developing the low-cost and high-powered supercapacitors [2]. Furthermore, compositing the nanostructured TMOs with carbon-based materials could remarkably improve the SC properties of TMOs [3]. In this regard, it has been clearly verified that the metal oxides like as nickel oxide [4-6], cobalt oxide [7-9], iron oxide [10-15] and manganese oxide [16-19] are promising materials for use in SCs as compared with other TMOs. Furthermore, manganese dioxides (MnO_2 and Mn_3O_4) have attracted much attention for its good electrochemical capacitive properties, wide potential range, low cost, and environmental friendliness [20,21]. However, the low electrical conductivity of manganese oxides (10^{-5} – 10^{-6} S/cm) lead to its poor electrochemical capability in energy storage applications like batteries and capacitors [22]. It has been reported that wrapping or decoration of manganese oxide onto a conductive substrates could overcome the mentioned problem to a certain extent. Until now, various conductive networks including activated carbon [23-25], carbon particles [26], carbon nanotubes [27-29], carbon fiber [30,31], nickel foam [32], metal ion doping [33], graphene [34-38], graphene oxide [39,40] and reduced GO [41-46] have been applied to load manganese oxide, which has been resulted an increasing conductivity and restricting volume change during GCD process. Combining manganese oxides with graphene-based networks could improve the electrical conductivity and stability of the fabricated electro-active materials, and enhance its electrochemical performances [34-43]. Among the graphene-based conductive material used for preparation of manganese oxide composites, reduced graphene oxide sheet (rGO) is reported to be a more suitable one as compared with conductive polymers due to its prominent advantages of high conductivity and stability [44-46]. However, the mass loading of oxide material onto the carbon-based network is still major problem. In some work, layer by layer deposition of composite component has been proposed and it has been observed that the low conductivity problem could be effectively refined [47,48]. Herein we developed an electrochemical/electrophoretic approach for simultaneously formation of rGO/manganese oxide composite, where the highest contact between carbon-based network and oxide part of composite could be provided and the excellent electrochemical performance is expected. The cathodic electrochemical synthesis strategy is used for decoration of manganese oxide onto rGo sheets. Cathodic electrochemical synthesis has been considered as powerful method for synthesis of various TMOs [49-59]. However, this route has been rarely used for fabrication of TMOs/graphene composites [60-62]. Electrophoretic deposition (EPD) is also applied as a facile route for composite synthesis [63,64]. Here, we used co-deposition of GO and manganese oxide through EPD and ECD,

respectively, for the fabrication of rGO-Mn₃O₄ nanocomposite. To the best of our knowledge, preparation of rGO-Mn₃O₄ composite has not been reported through this method. The crystal phase, chemical structure, morphology and electrochemical properties of the prepared composite were investigated by XRD, IR, FE-SEM, TEM, cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) techniques. The results verified the successful preparation of the mentioned composite *via* simultaneous EPD-ECD strategy.

2. EXPERIMENTAL PROCEDURE

2.1. Materials

Mn nitrate tetrahydrate (Mn(NO₃)₂·4H₂O, 99.5%), Na₂SO₄ were purchased from Sigma Aldrich. Graphene oxide (GO) was provided from Research Institute of Petroleum Industry (IRIP). All materials were used as received, without any purification.

2.2. Synthesis of rGO-Mn₃O₄ composite

For the synthesis of Mn₃O₄/rGO composite, electrophoretic deposition/electrochemical deposition techniques were used. Electrochemical set up includes an electrochemical cell containing deposition electrolyte and a home-made electrochemical workstation system (Potentiostat/Galvanostat, Model: NCF-PGS 2012, Iran). The cell electrodes were a steel sheet (surface area of 1 cm²) centered between two parallel graphite anodes. The deposition time and bath temperature were 20 min and 25 °C, respectively. An aqueous electrolyte was prepared as follow; First, 0.2 g graphene oxide powder was dispersed in 100 cc deionized water by assisted of probe sonicator. Then, 0.8 g manganese nitrate was added into graphene oxide dispersed solution and stirred for 30 min. This prepared electrolyte was used as the electrodeposition bath in the synthesis of rGO-Mn₃O₄ composite. To deposition of the rGO-Mn₃O₄ composite onto steel grid, current density of 5 mA cm⁻² was applied into the prepared electrolyte for 30min. After deposition, the steel sheet was washed several times with the deionized H₂O and dried at 70 °C for 1h. The fabricated electrode (rGO-Mn₃O₄@steel sheet) was directly used as working electrode in the capacitance measurement tests. Furthermore, the deposited composite film was scrapped and used in powder form for structural characterization through XRD, IR, SEM and TEM analyses.

2.3. Instruments

Morphologies of the synthesized composite powder were observed through field-emission scanning electron microscopy (FE-SEM, Mira 3-XMU with accelerating voltage of 100 kV). The crystal phase of the prepared composite was specified by X-ray diffraction (XRD, Phillips PW-1800) using a Co K α radiation. Transmission electron microscopy (TEM, model

Zeiss EM900 with an accelerating voltage of 80 kV) was used to provide high resolution images. The chemical bonds in the prepared composite were studied through FTIR spectroscopy using model Bruker Vector 22 FT-IR spectrometer.

2.4. Electrochemical evaluations

The electrochemical evaluations were carried out on a AUTOLAB® (Eco Chemie, PGSTAT 30) electrochemical analyzer in a three-electrode system. The working electrode was the synthesized rGO-Mn₃O₄@steel sheet with mass loading of 3 mg. A conventional three-electrode system, containing the working electrode, Pt counter electrode and Ag/AgCl reference electrode was assembled to perform electrochemical evaluations. First, electrochemical experiments were done in a conventional three-electrode cell with 1 M Na₂SO₄ electrolyte. The capacitance performance of the prepared composite electrode was characterized by using cyclic voltammetry (CV) and galvanostatic charge-discharge techniques. The CVs of the fabricated composite electrode were performed in the potential window of 0 to +0.9 V vs. Ag/AgCl at the potential sweeps of 5, 10, 25, 50 and 100 mV s⁻¹. The charge-discharge diagrams were recorded at the different currents of 0.5, 1, 2, 3 and 5 A g⁻¹ within a potential range of 0 to +0.9V vs. Ag/AgCl. The specific capacitances (Cs) of the fabricated composite were calculated from the applied techniques i.e. from Eq. (1) using CV data and from Eq. (2) using GCD data [65]:

$$Cs(F g^{-1}) = \frac{Q}{m\Delta(V)}, \quad Q = \int_{V_a}^{V_a} I(V)dV \quad (1)$$

Where Cs is the specific capacitance (F/g), *m* is the mass of the electrode (g), *v* is the scan rate of CV curves (V/s), and (Va– Vc) represents the potential window (V) [66].

$$Cs (F g^{-1}) = \frac{Q}{m \times \Delta V}, \quad Q = I \times \Delta t \quad (2)$$

Where Cs is the specific capacitance (F/g), I is the current load (A), *m* is the composite mass load onto the steel sheet (g), ΔV is the applied potential range (V), and Δt is the discharge time (s).

3. RESULTS AND DISCUSSION

3.1. Structural and morphological characterizations

The X-ray diffraction patterns of the graphene oxide powder and the electro-synthesized composite is presented in Fig. 1. The characteristic diffraction peak of GO is observed at the 2theta of 9.5 degree (Fig. 1a). In the pattern of composite (Fig. 1b), all the observed diffraction peaks are correspond to the crystalline hausmannite (Mn₃O₄) (JCPDS 01-1127)

[67]. Moreover, the peak for the graphene oxide sheets is almost disappeared, suggesting that the rGO sheets in the rGO-Mn₃O₄ nanocomposite are stacked in a disordered manner [68]. These results indicate that the fabricated rGO-Mn₃O₄ nanocomposite is composed of Mn₃O₄ nanoparticles and disorderedly stacked rGO nanosheets.

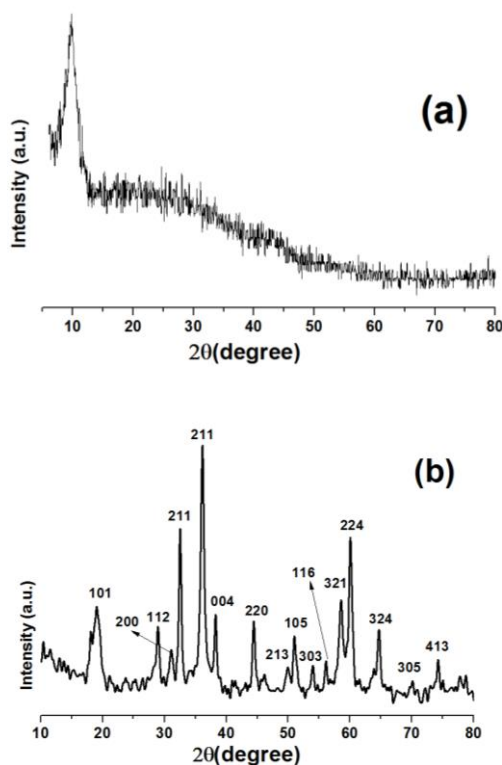


Fig. 1. XRD patterns of (a) GO powder and (b) the electro-synthesized rGO-Mn₃O₄ composite

FT-IR spectra of the GO and its composite with manganese oxide are presented in Fig. 2. In the spectrum of GO, a strong and broad absorption at 3429 cm⁻¹ due to O–H stretching vibration of carboxyl groups and the absorbed water. The peaks at 1403 and 1274 cm⁻¹ correspond to the skeletal vibrations of C–OH and C–O–C [42,69]. The characteristic peak for C=O stretching vibration appears at 1723 cm⁻¹[69,44]. The peak at 1632 cm⁻¹ corresponds to C=C skeletal stretching vibration. The stretching peak appears at 2922 cm⁻¹ and 1462 cm⁻¹ are attributable to C–H and C–C band, which refer to skeletal vibrations of non-oxidized graphitic domains [42]. The presence of peak at 1588 cm⁻¹ belong to C=O group band could be assigned to the unreduced –C=OH group. The peak at 1061 cm⁻¹ suggested C–O vibrations of carboxylic acid [44]. In the spectrum of the composite (Fig. 2b), the new IR bands (at 490 cm⁻¹ and 620 cm⁻¹) are observed, which are due to Mn–O stretching mode in tetrahedral and octahedral sites [42,44]. Furthermore, the absorption band at 3459 cm⁻¹ is due to the O–H vibrations from adsorbed moisture. Notably, the peak relating to the C=O groups is not appeared in FTIR spectrum of the composite nanocomposites, confirming that GO is reduced

to rGO during the electrochemical deposition of Mn_3O_4 . These results verified the successful synthesis of rGO- Mn_3O_4 composite through EPD-ECD method.

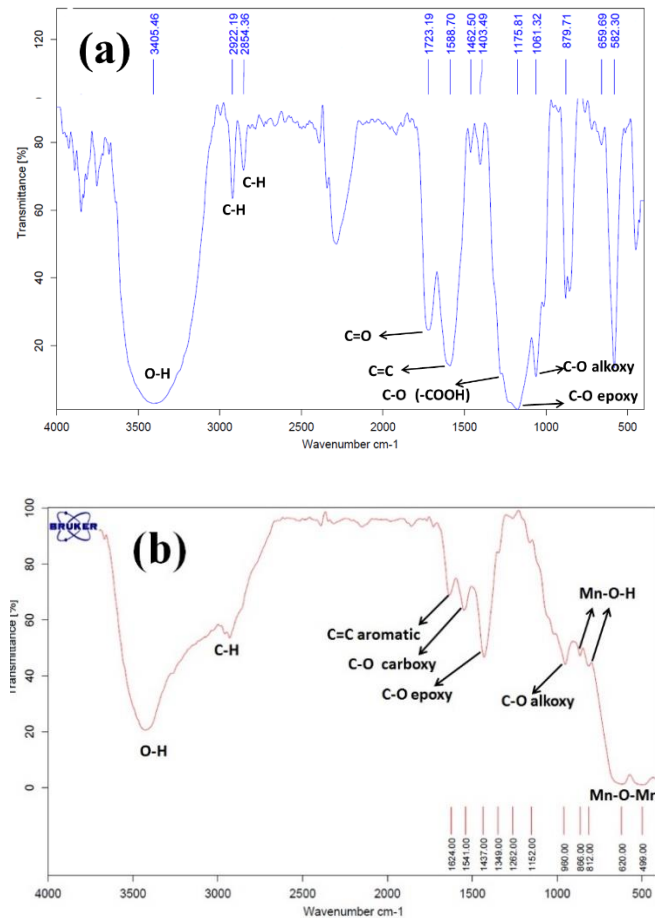


Fig. 2. IR spectra of (a) GO and (b) rGO- Mn_3O_4 composite

The morphology of the fabricated composite is given in Figs. 3a and b. For the prepared composite, two types of morphologies (i.e. particle and sheet) are seen in the FE-SEM images. The sheet-like texture is assigned to the reduced graphene oxide part of the composite and the particle one is indicate the electrochemically grown Mn_3O_4 deposits onto the rGO sheets (Fig. 3a). In fact, these observations suggested that Mn_3O_4 nanoparticles are intimately anchored on the rGO sheets. Furthermore, the surfaces of rGO sheets are fully filled by Mn_3O_4 nanoparticles, as shown in Fig. 3b. The energy-dispersive X-ray (EDS) data of the composite is also given in Fig. 3c. The presence of 19.68%wt Mn, 64.25%wt O, and 16.08%wt C elements are observed, which fully proved the Mn_3O_4 and rGO components in the fabricated composite. As observed in Fig. 3b, the oxide particles are decorated onto the rGO sheets. This fact suggested that the electrochemical grown of manganese oxide NPs is occurred on the electrophoretically deposited rGO sheets onto the cathode surface. Hence, an

EPD-ECD mechanism was proposed for the formation of rGO-Mn₃O₄ composite on the cathode electrode. The schematic view of the proposed mechanism is presented in Fig. 4.

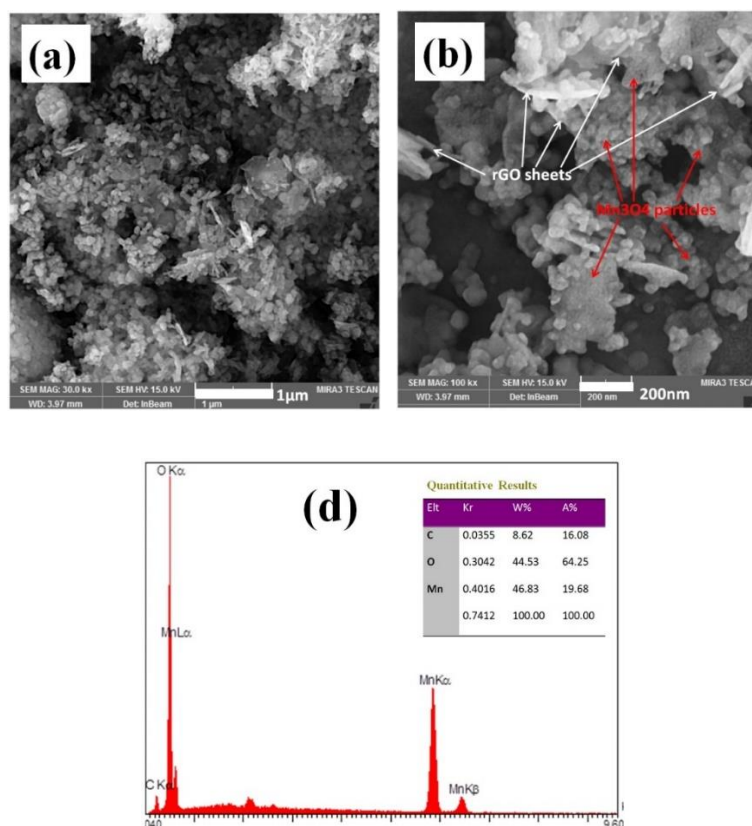


Fig. 3. (a,b) FE-SEM images the electro-synthesized rGO-Mn₃O₄ composite and (c) its EDS data

First, GO sheets dispersed in water electrolyte through negatively charged carboxylic groups are electrophoretically deposited into the cathode surface as shown in Fig. 4a. Notably, the metal ions (Mn²⁺) connected to the negatively charged COO⁻ groups of GO are also moved to the cathode surface together with the GO sheets, as seen in Fig. 4a. As the GO sheet receive to the cathode surface, the formation and deposition Mn₃O₄ particle is occurred on the cathode surface (Fig. 4b). Furthermore, the reduction of GO is also taken placed at the high basic condition on the cathode surface. In final, composition of rGO-Fe₃O₄ nanoparticles are formed on the cathode (Fig. 4b), which are completely verified through TEM observations (Fig. 5). For the electrodeposited pristine iron oxide, TEM observations revealed particle morphology with average diameter of 10 nm (Fig. 5a). For the composite, well dispersed thin rGO sheets are clearly observed where iron oxide particles (diameter≈15 nm) are present on their surfaces (Fig. 5b). This arrangement of oxide NPs onto rGO sheet prevents their agglomeration since iron oxide NPs tend to stick each other as a result of their magnetic character, where the agglomerated particles were observed for the pristine Fe₃O₄

NPs (Fig. 5a). TEM images in Fig. 5 indicated that the Mn_3O_4 NPs (with sizes of 10 nm) have been formed onto the rGO sheets.

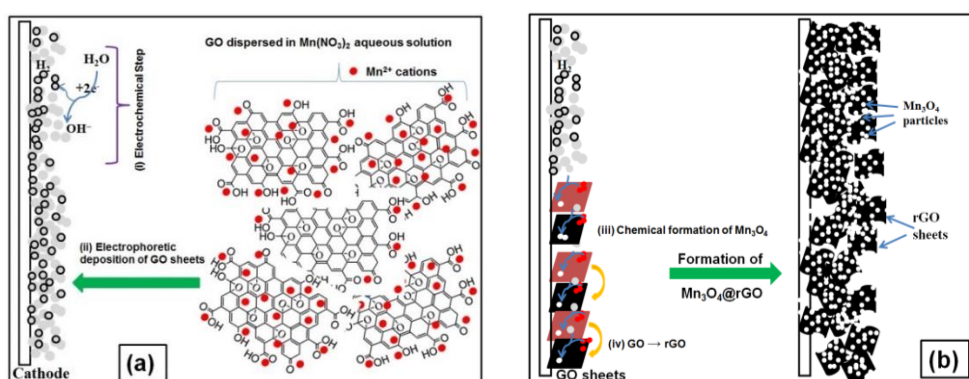


Fig. 4. Formation mechanism of rGO-Mn₃O₄ composite

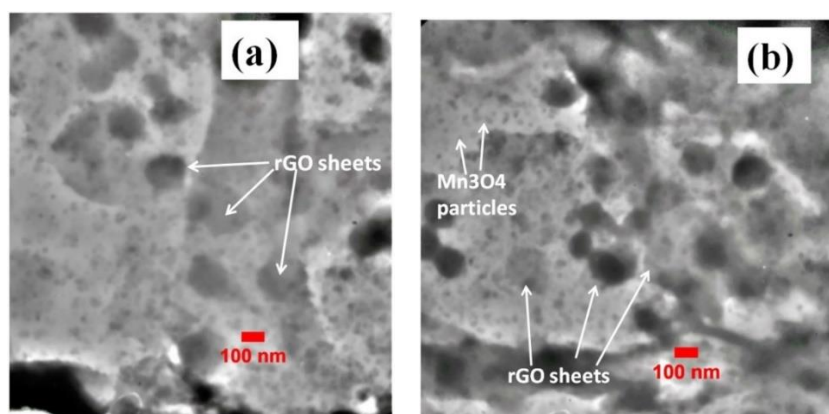


Fig. 5. TEM images of the prepared GO-Mn₃O₄ composite

3.2. Capacitance measurements

The electrochemical performance of the fabricated rGO-Mn₃O₄ nanocomposite was first studied using cycle voltammetry in 1 M Na₂SO₄ electrolyte at a potential window of 0-0.9 V. As seen in Fig. 6a, the fabricated rGO-Mn₃O₄ electrode have nearly rectangle-shaped and symmetric CV curves without any redox peak currents at all the applied scan rates, suggesting the ideal capacitive nature [43,44]. The CV current response of the fabricated increases gradually with the increase of scan rate and no significant change in the shape of CV curve even at high scan rates, indicating an excellent capacitance behavior and the fast diffusion of electrolyte ions into the fabricated rGO-Mn₃O₄ electrode. Using Eq. (1), the capacitances at 5, 10, 25, 50 and 100 mV/s were calculated to be 454, 419, 378, 344 and 302 F/g, respectively.

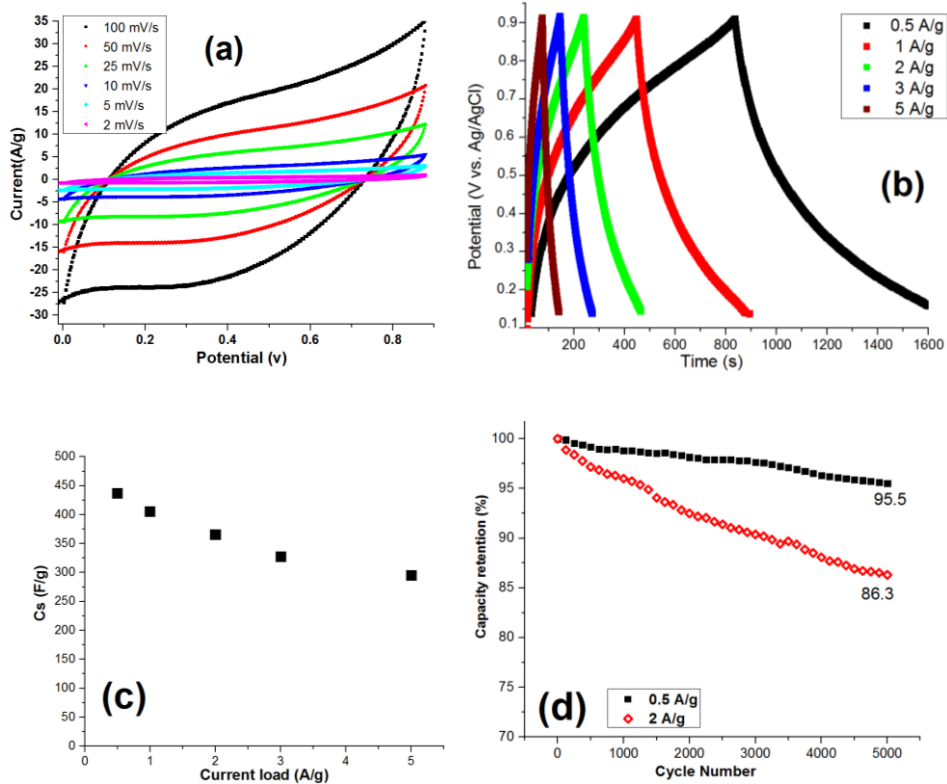


Fig. 6. (a) cyclic voltammograms; (b) charge-discharge; (c) the calculated capacitance values and (d) cycling stability for the fabricated rGO-Mn₃O₄ electrode

Fig. 6b indicates the charge-discharge profiles of the fabricated electrode at various current densities over the potential range of 0-0.9 V. No obviously shape change was observed among the nearly linear GCD curves at all applied current loads, implicating the good rate capability and reversibility. The charge/discharge time varies at a constant current density for Mn₃O₄/rGO electrode, suggesting different Cs values. As shown in Fig. 6c, the Cs values of rGO-Mn₃O₄ nanocomposite electrode were calculated to be 438, 405, 364, 329 and 297 F/g at 0.5, 1, 2, 3 and 5 A/g, respectively, which is higher than the reported Cs values for pristine nanostructured Mn₃O₄ and its composite electrode in the literature; e.g. 210 F/g for nanostructured Mn₃O₄ at 0.5 A/g [16], 298 F/g for Mn₃O₄ nanorods at 1 A/g [17], 279 F/g for vertically-aligned Mn₃O₄ nanorods at 1 A/g [19], 293 F/g for Mn₃O₄ nanorods/Ni foam at 0.5 A/g [32], 272 F/g for Cr-doped Mn₃O₄ nanocrystals at 0.5 A/g [33], 258.6 F/g for Mn₃O₄/rGO nanocomposite at 0.25 A/g [41], 342.5 F/g for Mn₃O₄/rGO electrode at 1 A/g [42], 125 F/g for rGO/Mn₃O₄ composite at 1 A/g [44], 457 F/g for rGO-Mn₃O₄ at 1 A/g [45], 207.2 F/g for Mn₃O₄/rGO at 1 A/g [46], and 194.5 F/g for Mn₃O₄/RGO-0.35 electrode at 0.5 A/g [70]. These enhancements signified the positive synergistic corporation between reduced graphene oxide and hausmannite which may originate from the following issues: (a) uniformly anchoring the Mn₃O₄ particles on reduced graphene oxide sheets (as confirmed by TEM observations in Fig. 5), (b) reduced graphene oxide sheets provides the highly conductive electronic network for the redox process of electrode, (c) incorporation of Mn₃O₄

particles into the reduced graphene oxide sheets could finish the re-stacking of reduced graphene oxide sheets, and (d) reduced graphene oxide could prevent the volume change and agglomeration of oxide during GDC process. These events maximize the following cases; (i) interface area between Mn_3O_4 particles and reduced graphene oxide sheets, (ii) the electrochemical usage of electro-active materials, and (iii) ionic, electrical and charge transfer pathways. The above mentioned characters of rGO- Mn_3O_4 nanocomposite give it to exhibit the observed supercapacitive performances.

The fabricated composite electrode was cycled 5000 times at the currents of 0.5 and 2 A g^{-1} , and the calculated capacity retention values are shown in Fig. 5d. It was observed that the fabricated electrode could deliver 95.5% and 86.3% of its initial capacity after 5000 times cycling at current of 0.5 and 2 A g^{-1} . These data specified the suitable cycling performance of the composite.

4. CONCLUSION

In summary, rGO- Mn_3O_4 nanocomposite was successfully synthesized through a mixed electrophoretic/electrochemical deposition approach. Highly dispersed Mn_3O_4 nanoparticles with a size of 10 nm were in situ deposited on the surface of rGO sheets with uniform particle distribution. The prepared rGO- Mn_3O_4 nanocomposite were used as electrode materials for supercapacitor. It was observed that the fabricated rGO- Mn_3O_4 nanocomposite is able to deliver greatly enhanced capacitive performance and long-term stability. Based on the obtained results, the one-pot EPD/ECD procedure is proposed as facile synthesis method for fabrication of binder-free oxide/reduced graphene electrodes for supercapacitor uses.

REFERENCES

- [1] F. Wang, X. Wu, X. Yuan, Z. Liu, Y. Zhang, L. Fu, Y. Zhu, Q. Zhou, Y. Wu, and W. H. Chem. Soc. Rev. 46 (2017) 6816.
- [2] D. P. Dubal, O. Ayyad, V. Ruiz, and P. Gómez-Romero Chem. Soc. Rev. 44 (2015) 1777.
- [3] G. Zhang, X. Xiao, B. Li, P. Gu, H. Xue, and H. Pang, J. Mater. Chem. A 5 (2017) 8155.
- [4] M. Aghazadeh, J. Mater. Sci.: Mater. Electron. 28 (2016) 3108.
- [5] M. Aghazadeh, and M. R. Ganjali, J. Mater. Sci.: Mater. Electron. 28 (2017) 8144.
- [6] M. Aghazadeh, A. Rashidi, and M. R. Ganjali, Int. J. Electrochem. Sci. 11 (2016) 11002.
- [7] M. Aghazadeh, J. Appl. Electrochem. 42 (2012) 89.
- [8] M. Aghazadeh, R. Ahmadi, D. Gharailou, M. R. Ganjali, and P. Norouzi, J. Mater. Sci.: Mater. Electron. 27 (2016) 8623.

- [9] M. Aghazadeh, M. Hosseini-fard, B. Sabour, and S. Dalvand, *Appl. Surf. Sci.* 287 (2013) 187.
- [10] M. Aghazadeh, I. Karimzadeh, and M. R. Ganjali, *J. Mater. Sci.: Mater. Electron.* 28 (2017) 13532.
- [11] M. Aghazadeh, *J. Mater. Sci.: Mater. Electron.* 28 (2017) 18755.
- [12] M. Aghazadeh, and M. R. Ganjali, *J. Mater. Sci.* 53 (2018) 295.
- [13] M. Aghazadeh, I. Karimzadeh, M. R. Ganjali, and A. Behzad, *J. Mater. Sci.: Mater. Electron.* 28 (2017) 18121.
- [14] M. Aghazadeh, and M. R. Ganjali, *Ceram. Int.* 44 (2018) 520.
- [15] M. Aghazadeh, I. Karimzadeh, M. Ghannadi Maragheh, and M. R. Ganjali, *Korean J. Chem. Engin.* 35 (2018) 1341.
- [16] Z. Qi, A. Younis, D. Chu, and S. Li, *Nano-Micro Lett.* 8 (2016) 165.
- [17] M. Aghazadeh, A. Bahrami-Samani, D. Gharailou, M.G. Maragheh, and M. R. Ganjali, *J. Mater. Sci.: Mater. Electron.* 27 (2016) 11192.
- [18] M. Aghazadeh, M. R. Ganjali, and P. Norouzi, *J. Mater. Sci.: Mater. Electron.* 27 (2016) 7707.
- [19] M. Aghazadeh, M. R. Ganjali, and P. Noruzi, *Thin Solid Films* 634 (2017) 24.
- [20] W. F. Wei, X. W. Cui, W. X. Chen, and D. G. Ivey, *Chem. Soc. Rev.* 40 (2011) 1697.
- [21] T. Kou, B. Yao, T. Liu, and Y. Li, *J. Mater. Chem. A* 5 (2017) 17151.
- [22] T. Zhai, X. Lu, F. Wang, H. Xia, and Y. Tong, *Nanoscale Horiz.* 1 (2016) 109.
- [23] X. Xiao, Y. Wang, G. Chen, L. Wang, and Y. Wang, *J. Alloys Compd.* 703 (2017) 163.
- [24] C. Liu, H. Song, C. Zhang, Y. Liu, C. Zhang, X. Nan, and G. Cao, *Nano Res.* 8 (2015) 3372.
- [25] L. Xu, M. Jia, Y. Li, X. Jin, and F. Zhang, *Sci. Rep.* 7 (2017) 12857.
- [26] S. Nagamuthu, S. Vijayakumar, and G. Muralidharan, *Energy Fuels* 27 (2013) 3508.
- [27] Y. Z. Song, R. X. Zhao, K. K. Zhang, J. J. Ding, X. M. Lv, M. Chen, and J. M. Xie, *Electrochim. Acta* 230 (2017) 350.
- [28] M. Mandal, D. Ghosh, K. Chattopadhyay, and C. K. Das, *J. Electron. Mater.* 45 (2016) 491.
- [29] H. Z. Chi, S. Tian, X. Hu, H. Qin, and J. Xia, *J. Alloys Compd.* 587 (2014) 354.
- [30] H. Cheng, H. M. Duong, and D. Jewell, *RSC Adv.* 6 (2016) 36954.
- [31] H. Wang, C. Xu, Y. Chen, and Y. Wang, *Energy Storage Mater.* 8 (2017) 127.
- [32] D. Li, F. Meng, X. Yan, L. Yang, H. Heng, and Y. Zhu, *Nanoscale Res. Lett.* 8 (2013) 535.
- [33] R. Dong, Q. Ye, L. Kuang, X. Lu, Y. Zhang, X. Zhang, G. Tan, Y. Wen, and F. Wang, *ACS Appl. Mater. Interfaces* 5 (2013) 9508.
- [34] Y. Tian, D. Li, J. Liu, H. Wang, J. Zhang, Y. Zheng, T. Liu, and S. Hou, *Electrochim. Acta* 257 (2017) 155.

- [35] A. E. Rashedab, A. A. El-Moneim, *Mater. Today energy* 3 (2017) 24.
- [36] X. Qiu, D. Xu, L. Ma, and Y. Wang, *Int. J. Electrochem. Sci.* 12 (2017) 2173.
- [37] S. Hassan, M. Suzuki, and A. Abd El-Moneim, *Int. J. Electrochem. Sci.* 9 (2014) 8340.
- [38] C. Xiong, T. Li, M. Khan, H. Li, and T. Zhao, *RSC Adv.* 5 (2015) 85613.
- [39] Y. Wang, *J. Mater. Res.* 30 (2015) 4.
- [40] F. Ding, N. Zhang, C. Zhang, and C. Zhang, *Mater. Transactions* 56 (2015) 1857.
- [41] J. Xu, X. Fan, Q. Xia, Z. Shao, B. Pei, Z. Yang, Z. Chen, and W. Zhang, *J. Alloys Compd.* 685 (2016) 949.
- [42] Y. Zhou, L. Guo, W. Shi, X. Zou, B. Xiang, and S. Xing, *Mater.* 11 (2018) 881.
- [43] T. Xiong, W. Siang, V. Lee, X. Huang, and J. M. Xue, *J. Mater. Chem. A* 5 (2017) 12762.
- [44] X. She, X. Zhang, J. Liu, L. Lia, X. Yu, Z. Huang, and S. Shang, *Mater. Res. Bull.* 70 (2015) 945.
- [45] H. U. Shaha, F. Wang, M. Sufyan Javed, N. Shaheen, M. Saleemd, and Y. Lia, *Ceram. Int.* 44 (2018) 3580.
- [46] P. Rosaiah, J. Zhu, D. P. M. D. Shaik, O. M. Hussain, Y. Qiu, and L. Zhao, *J. Electroanal. Chem.* 794 (2017) 78.
- [47] H. Jiang, L. Yang, C. Li, C. Yan, P. S. Lee, and J. Ma, *Energy Environ. Sci.* 4 (2011) 1813.
- [48] J. Shin, D. Shin, H. Hwang, T. Yeo, S. Park, and W. Choi, *J. Mater. Chem. A* 5 (2017) 13488.
- [49] M. Aghazadeh, A. N. Golikand, M. Ghaemi, and T. Yousefi, *J. Electrochem. Soc.* 158 (2011) E136.
- [50] T. Yousefi, A. N. Golikand, M. H. Mashhadizadeh, and M. Aghazadeh, *J. Taiwan Institute Chem. Eng.* 43 (2012) 614.
- [51] M. Aghazadeh, *Mater. Lett.* 211 (2018) 225.
- [52] M. Aghazadeh, M. G. Maragheh, M. R. Ganjali, and P. Norouzi, *RSC Adv.* 6 (2016) 10442.
- [53] M. Aghazadeh, A. Nozad, H. Adelkhani, and M. Ghaemi, *J. Electrochem. Soc.* 157 (2010) D519.
- [54] M. Aghazadeh, M. Ghaemi, A. N. Golikand, and A. Ahmadi, *Mater. Lett.* 65 (2011) 2545.
- [55] I. Karimzadeh, H. R. Dizaji, and M. Aghazadeh, *J. Magn. Magn. Mater.* 416 (2016) 81.
- [56] M. Aghazadeh, A. A. M. Barmi, H. M. Shirri, and S. Sedaghat, *Ceram. Int.* 39 (2013) 1045.
- [57] M. Aghazadeh, A. A. M. Barmi, and M. Hosseinifard, *Mater. Lett.* 73 (2012) 28.
- [58] M. Aghazadeh, M. G. Maragheh, M. R. Ganjali, and P. Norouzi, *Inorg. Nano-Metal Chem.* 27 (2017) 1085.

- [59] F. Khosrow-pour, M. Aghazadeh, and B. Arhami, *J. Electrochem. Soc.* 160 (2013) D150.
- [60] Rusi, and S. R. Majid, *Scientific Reports* 5 (2015) 16195.
- [61] S. Ghasemi, R. Hosseinzadeh, and M. Jafari, *Int. J. Hydrogen Energy* 40 (2015) 1037.
- [62] Y. Zhao, D. Zhao, P. Tang, Y. Wang, C. Xu, and H. Li, *Mater. Lett.* 76 (2012) 127.
- [63] A. Chavez-Valdez, M. S. P. Shaffer, and A. R. Boccaccini, *J. Phys. Chem. B* 117 (2013) 1502.
- [64] M. Diba, D. W. H. Fam, A. R. Boccaccini, and M. S. P. Shaffer, *Prog. Mater. Sci.* 82 (2016) 83.
- [65] M. Aghazadeh, and M. R. Ganjali, *J. Mater. Sci.: Mater. Electron.* 28 (2017) 11406.
- [66] M. Aghazadeh, I. Karimzadeh, and M. R. Ganjali, *Mater. Lett.* 209 (2017) 450.
- [67] R. Cheraghali, and M. Aghazadeh, *Anal. Bioanal. Electrochem* 8 (2016) 193.
- [68] S. Chen, J. W. Zhu, X. D. Wu, Q. F. Han, and X. Wang, *ACS Nano* 4 (2010) 2822.
- [69] T. Wang, Y. Li, S. Geng, C. Zhou, X. Jia, F. Yang, L. Zhang, X. Ren, and H. Yang, *RSC Adv.* 5 (2015) 88958.
- [70] Y. Wang, Z. Ji, X. Shen, K. Xu, and A. Yuan, *J. Alloys Compd.* 684 (2016) 366.